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MATERIALS NEEDS FOR COMPACT FUSION REACTORS

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The aconomic prospects for magnetic fusion energy can be dramatically improved if for the same total power output the fusion neutron first-well (FW) loading and the eystem power density can be increased by factors of 3-5 and 10-30, respectively. A number of "compact" fusion reactor ambodiments have been proposed, all of which would operate with increased FW loadings, would use thin (0.5-0.6 m) blankets, and would confine quari-steady-state pleams with resistive, water-cooled copper or aluminum coils. Increased system power density (5-15 MW/m³ versus 0.3-0.5 MW/m³), considerably reduced physical size of the fusion power core (FPC), and appreciably reduced economic laverage exerted by the FPC and associated physics result. The unique materials requirements enticipated for these compact reactors are outlined against the well documented backdrop provided by similar needs for the mainline approaches. Surprisingly, no single materials need that is unique to the compact systems is identified; crucial uncertainties for the compact approaches must also be addressed by the mainline approaches, particularly for in-vacuum componente (FWe, limiters, divertors, etc.).

1. INTRODUCTION

Both the technical and commercial success of magnetic fueton depend on edvences in engineering materials operating in environment of highly non-uniform surface and volumetric power densities. These heat loads will be applied under conditions where the basic engineering material properties of stressed components are being drematically eltered by an intense neutron/gremmerey/charged-particle irradiation field. The interdependence between pleams enginaering, reactor design, and materials wcience/engineering needed economic, commercially attractive fusion power has been highlighted by a number of excellent over lew papers dealing with first wells! (FW), blankets² (B), materials needs for epecific devices, 3,4 and the worldwide materials programs addressing these needs. 5, 6

Nygren³ pointe out that whese materials needs have been identified primarily by conceptual design studies, with the more exacting "designs to construct" eventually requiring

difficult materials choices, an expanded materials data base, considerably more design detail, and improved estimates of major ambsystem performance. Evan at the conceptual design level, however, the list of materials performance requirements presents a major challenge for the INTOR/DEMO/COMMERCIAL development sequence. The more compact, higher-power-density fusion approaches propose smaller fusion power cores (FPC, i.e., firstwell/blenket/ehield/coils) operating with incressed power density and FW neutron wnd heating loads. The degree to which materials performance requirements are altered by the needs of these compact fusion reactors is addressed qualitatively herein. The rationale, pathway, and generic technology required for the compact reactors have been described recently. 7, 8

After summarizing the reasons for considering systems with material requirements that in some cases may exceed those prespected in Refs. 1-4, the generic needs of compact devices are described. Specific compact

reactor designs have been suggested ⁸ for the Reversed-field Pinch (RFP), the Ohmically-Hested Toroidel Experiment (OHTE, an RFP with suxiliary helical windings), and the high-field tokemak. Other candidates for compact reactors have also been identified. ^{8, 9}

Although the matericle issues and needs addressed herein are generic, specific quantitative examples are referred to conceptual design results emerging for the compact RFP resector (CRFPR). 10 Similarly, comparisons with the mainline development sequence are made with the STARFIRE 11 and Culham MkIIB 12 tokemak resector designs.

2. COMPACT FUSION REACTORS

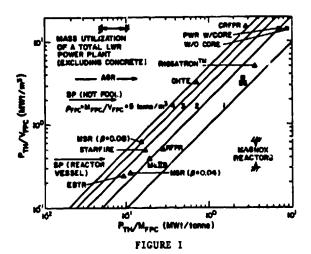
The dominance in mass and cost of the FPC for most approaches to magnetic fusion 7 has created interest in more compact, higher-power-density systems. The following improved characteristics are being pursued through the compact reactor option.

- FPC mass and volums comparable to alternative nuclear power systems (system power density of 5-15 MWt/m³, mass utilization of 0.3-0.5 tonns/MWt), which are fact a of 10-30 times better than values being projected for most magnetic fusion achames.
- e Reduced sensitivity of unit direct cost (UDC, \$/kWs) to the cost of the reactor plant equipment (RPE/TDC ≤ 0.3 rather than 0.5-0.8, where TDC is the total direct cost).
- Competitive eyerem costs and cost of electricity (COE, Bills/kWah) using reslictic unit materials costs, febrication/construction rises, and development achedules/costs.
- Rapid deployment of small FPCs with the potential for "block" installation and maintenance (i.e., single or few piece FPC), using systems relying on a minimum

extension of technology (e.g., resistive rather than superconducting coils, obmic hasting rather than high-frequency rf heating or neutral-beam injection, etc.).

This prescription for economically competitive fusion is not without risks or trade-offs; 7,8 potential for increased recirculating power, reduced thermal conversion efficiency, and reduced plant factor could leed to reduced plent efficiency, increased plent cost, and increased COE. Minimization of these risks will depend on the availability and use of materials and material engineering approaches that differ somewhat from those being suggested and pursued by the mainline programs. These differences ere highlighted herein.

Although hauristic arguments can be made to point the way towards improved system economics through higher eyetem power density or lover FPC mass utilization, ultimately deteiled peremetric studies on specific concepts must satablish economically optimum, technologically fessible systems. 10 For the present purposes, however, Fig. 1 continues with the heuristic approach by displaying the system power density versus the inverse of the FPC mess utilization; lines of unit slope on Fig. 1 give the everage FPC mass density, PPPC (tonne/m3). The eyetem power density for most of the "superconducting" fusion systems dim: played on Fig. 1 are at least one order of magnitude below other nuclear power systems. In order to gain an order of magnitude increase in this imports a parameter, an increese in FW neutron current by 3-5, simultaneously with a decrease by -2-3 in FW radius, B/S thickness, and coil radius and eize, ie required. 7 The former change makes steinless steel even less attractive from the hest-transfer viewpoint, whereas the reduced B/6 thickness sliminates superconducting coils



Comparison of system power densities being projected for conceptual fusion reactors with a number of fission reactor systems. STARFIRE tokamak (Ref. 11), Culham MkIIB tokamak (Ref. 12), Reversed-field Pinch Superconducting Reactor RFPR (Ref. 13), Modular Stellerator Reactor MSR (Ref. 14), ELMO Bumpy Torus 15), Magnox Ges-Cooled Reactor EBTR (Ref. Reactor (Ref. 16), Super Phenix Liquid-Metal Fast-Breader Fission Reactor SP (Ref. 17), Advance Gas Resctor ACR (Ref. 18), Compact Reversed-Field Pinch Reactor CRFPR (Ref. 10), Ohmically-Heat Toroidal Experment Reactor OHTE (Ref. 19), High-Field Tokemak Reactor, (Ref. 20), Pressurized-Weter Riggetron fission Reactor PWR (Ref. 21), PWR Steam Generator Si; (Ref. 21).

from consideration, since neutron fluxes and heat deposition in the coils cannot be kept low in the space available. Hence, the compact systems that emerge (CRFPR, OHTE, Riggetron) use resistive copper-alloy coils with ceremic electrical insulation and generally provide only a thin (0.5-0.6 m) blanket between the FW and the high-radiation-flux, resistive coils. In ceruain instances, FW (Riggetron) or near-FW (OHTE) activaly-driven coils may be necessary.

The compact systems depicted on Fig. 1 would achieve DT ignition by Ohmic dissipation of toroidal pleams currents. Inferred, therefore, is some form of inductive current drive, at least for startup; each system in

principle is capable of burn extension by non-For those compact reactors inductive means. with plasma confinement depending in part (i.e., OHTE) or totally (i.e., Riggstron) on strong toroidel fields, the magnet coils may be highly atressed as well as presenting a potentially serious drain on the overall plant efficiency (i.e., incressed recirculating power, reduced thermal recovery efficiency, etc.) Generally, the high-hest-flux FWe and other in-vecuum component (IVC) surfaces, thin high-power-density blankets, and resistive exo-blanket (CRFPR) or neer-FW (OUTE. Riggetron) resistive coils largely define the differences in materials requirements between the compect and the other magnetic fusion approaches.

Generally, two crucial quantions must be answered before the economic attractiveness of compact approaches to fueion power can be fully substantiated.

- e can a plasma confinement schame based aither on a mainline, alternative, or a combination thereof be found that will stably confine plasma of the required power density while giving some assurance of long-pulsed or steady-ateta operation with a recirculating power fraction < 0.15-0.207
- given the plasma physics inferred from the lest issue, can all subelements of the FPC (i.e., IVC, blanket/shield, coils) be made to operate with an acceptable engineered lifetime, both in terms of real time (i.e., maintenance period) and fluence (i.e., total amount of energy generated per mass of FPC consumed)?

The first issue is not within the scope of this paper, but second-stability-region tokemake, RFP/OHTEs, and apherosake/field-reversed configurations provide exciting presential on both theoretical and experimental

grounds. The second question of FPC lifetime as summarized in Table I, is complex, and centers on the materials theme of this overview. Four major determinants of FPC lifetime are identified in Table I: i cotor operating conditions; FPC material properties; component geometry and constraints; and design and failure criteria. By applying similar design and failure criteria to all fusion approaches, and assuming negligible influence of rate on the effects of radiation in changing materials properties (i.e., a fluence affect) the FPC lifetime issue becomes one of reactor operations and component geometry.

Operating 1 n the compact regime eignificently influences both reactor operations and geometry (i.e., size). The major change in reactor operating conditions is the incressed heat/perticls fluxes, but designing to the same failure criteria should differences, albeit aliminata these potentially at a higher cost. The reactor operational flexibility afforded by smaller FPCe, perticularly with respect to the lest point in Table I listed under component geometry, potentially can offeat the edded cost of designing for a more highly atresped rescoor operating condition in order to essure that each unit mass of FPC delivers the economically necessary emount of energy within its lifetime. This issue of total ("batch") versus pertiel ("petch") FPC maintenance, elthough difficult to quantify, is best depicted on Fig. 2, which compares a compact reactor (the CRFPR, the OHTE reactor is of similer size) with both a PWR and the STARFIRE tokemak reactor. In summary, therefore, the key elements of the FPC lifetime issue (Table I) may either a) be common to fusion in general, or b) have a mutually salf-concesling impact [i.e., more severe reactor operation in a more favorable reactor geometry (size)].

TABLE I. SUMMARY OF FPC LIFETIME DETERMINANTS 22

- · Reactor Operating Conditions
 - FW neutron loading
 - -- Volumetric heating
 - -- damage rates (dps/yr, He appm/yr, H appm/yr, burnup)
 - Plasma energy rejection
 - -- particle fluxes to IVC (DT, neutrels, He, impurities)
 - -- heat fluxes (conduction and radiation)
 - Duty cycle
 - Coolent (kind and temperature/pressure)
- · Material Properties
 - thermal (heat capacity, conductivity, expensivity)
 - mechanical (Young's modulus, ultimate and yield atreases, uniform alongation, total alongation, fracture toughness, creep, fatigue, crack growth, awelling)
 - electrical (conductivity)
 - nuclear (alloying constituents, transmutations, gas production, dps, radioactivity, afterheat).
 - aurfaces (aputtering, adsorption, gas racycle, electron emission)
- Component Geometry and Constraint
- etrese end temperature distributions
- component interactions/interplay
- size and degree prachack/shakedown sllowed,
 QA, replacement/repair time
- · Design and Failure Criteria
 - electic deformation and electic instability
- pleatic deformation and pleatic inetability (incremental collapse/retchatting)
- brittle frecture
- stress rupture/creep deformation
- high-strain/low-cycle fetigue and creep/ fetigue interaction
- stress corrosion
- corrosion fatigus
- ewelling and differential volume change
- undesirable changes in material properties (ambrittlement, DBTT, electrical resistivity).

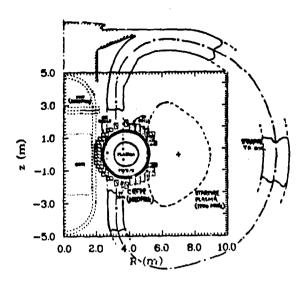


FIGURE 2
Cross sectional comparison of a compact fusion reactor design (CRFPR) with a fission reactor pressure vassel (PWR) and the STARFIRE tokamak reactor.

Although the scope of this overview does not allow a comprehensive assessment of the "compact" versue "conventional" ayetems, Table II nevertheless is included to give a quentitative example of the physics. angineering, and economic differences between comprehensive tokemak designs 11, 12, 23 and a compact RFP reactor design. 10 Since the demands on engineering materials performance are primarily generated by the thermal rediction and mechanical (stress, environment created by high-powerdensity plasms, FW, and blanket operation, the neutronice results 24 from a specific highpower-density FPC is given in Table III. This design elso use a 20-mm-thick copper-slloy FW, which allows some inferences to be made for those compact reactors requiring TW resistive coils. Acain. the comperisons quentitative information given in Figs. 1-2 and Tables It-III are intended to demonstrate the "order-of-magnitude" differences between the compact and more "conventional" approaches

to fusion power rather than to emphasize differences between specific conceptual reactor designs.

3. MATERIALS ISSUES/NEEDS

The key materials issues and needs for fusion in general can be divided according to the following three FPC subsystems:

- In-vacuum Components (IVC) 1
 - first wall
 - limiter
 - divertor
 - coile
 - ancennae
 - windows (rf)
- Blanket/Shield (B/S)²
 - breeder
 - coolant
 - etructure
 - multiplier
 - reflector/moderator
 - tritium berrier
 - ducts (rf, beems, fueling, vacuum, coolente)
- Magnet Coile (C)
 - conductor (superconductor versus resistive)
 - insulator (organic vareus inorganic)
 - structure
 - coolent (He(1) versus water)
 - kinds (TF, PF, OH, EF, active feedback, passive shell)

In addition to comprehensive materials needs essessments for these subsystems, 1-4 general reviews of fusion materials needs are available. 25 The technology needs for the compact systems have also been submarized recently. 8 No attempt is made here to repeat or to summarize these reviews and assessments. Instead, based on the general system differences and goals as outlined in Sec. 2. and Table II, differences in materials needs

TABLE II. PLASMA, COSTING AND FPC PARAMETER COMPARISON BETWEEN

FUSION-POWER-CORE (FPC) PERFORMANCE COMPARISONS(*)			
PARAMETER	STARFIEL 11	MK118 12-23	CRFFR)*
Gross thermal power, Pry (Mic)	4000	3261	3389
Sienker energy multiplication, Na	1.14	1-14	1.10
Thermal conversion efficiency, were	0.36	0.37	0.35
Recirculating power fraction, t	0.167	0.08	0.16
Place officiency, no = ngg (1-c)	0.30	0.34	0.30
Her electrical power, Fg(HNe)	1200.	à200.	1000.
Nom-mel 8/8 thickness Ab(m)	2.5	3.2	0.60
Mominal coli thickness, &c(m)	1.6	3.2	0.45
PPC volume, V _{PPC} (m ³)	8110(6630)	0000(4401)	242
First-well erez, A _w (m ²)	780	716	112
PPC volume/autface, VFFC/Au	10.4(8.50)	11.2(6.13)	2.16
System minor redius,			
$r_o = (V_{PPC}/2e^{2k_T})^{1/2}$ (m)	7.66(6.*0)	7.7.(3.77)	1.60
Planus chasher volume, V _{PC} (m ³)	1106(950)	870(836)	42.
P(rer-vell radius, ru(a)	2.83	2.71	0.73
PV neutron loading, $I_w(MH neutron/m^2)$	3.6	3.2	19.3
Power densisy, PTH/VPPC (MMt/m3)	0.50(0.66)	0.41(0.74)	14.
PPC mees, HppC (tonns)	23174/16496	17330 ^(b)	1160
• FV/8	1374	4700	223
• Shield	13360/6682	3630	
• Colle	8240	8980	937
Mass utilitation MpPC/PTM (tenne/MMt)	3.7/4.1	3.3	0.40
PPC denelsy, Mppc/Vppc (tonne/u3)	2.86/266	2.17(3.94)	3.6
Area den-ltp. pppclVppc/Appl(tonne/m2)	29.7	24.3	12.1
PPC come (H\$)	440.1/363.3	719-1(473-9)	43.6
• PW/S	82.4	204.3]13.50]	14.8
• Shield	186.1/109.3	137.2(90.3)	
• Colle	171-6	377.41248.61	30.8
PPC unit cost, c _{PPC} (\$/kg)	19.0/22.0	41.3127.31	37.0
PPC volumetric cost, cppc(M\$/m3)	0.033/0.039	0.16]0.11]	0.20
PPC eres tost, (PPC cost)/Ag(M\$/m2)	0.44/0.31	1.0010.44)	0.34
Cost Pigutes of Merit			
• RPE/TDC	0.36	0.72	0.36
• PPC/TDC	0.26/0.21	0.23	0.04
• (PW/1)/TDC	0.050	0.067	0.017

(a) Values in () based on toroidal volume, ashervice volume at co	et re l
toluen included. Values to right of / do see include vacuum po	ene i ng
ducts and posts. Values in \$/ convession in 1977 followed	
inflation face 1977 to 1980; otherwise the conversion/islistion	
le terereed:	

⁽b)Dose no: Inc ude 33,000 tonne l'en core.

between the mainline and the compact approaches are highlighted. Each of the three FPC major subsystems listed above is treated separately. Materials needs for subsystems outside the FPC are expected to be similar for all approaches and, therefore, are not discussed.

PLASMA PEYSICS/ENGINE RING JARAMETERS

PARAMETERS	STARFIRE!	MK11812.23	CRFFE! 0
Hajor radius, R _g (m)	7.C	6.7	5.79
Aspect ratio, A	3.6	5.5	5.3
Plasma elongation, c	1.4	1.75	1.0
Plasma triangularity, d	0.5	0.	0.
Average plasma miner fedius, r _p (u)	2.30	2.51	0.71
Places volume, T _p (m ²)	781.	836.	57.7
Average bets,	0.067	0-092	0.20
Magnetic field at plasms, S _o (T)	3.8	4.0	0.56(3.2)(4)
Safety fector ot limiter, q	5-1	2.5	0.02
Plosus turoidal carrent, 1 (NA)	10.1	10-2	18.4
Please current density, 10/HTp2(MA/m1)	0.54	0.31	11.8
Average electron temperature, To(keV)	17.3	12.0	20.0
Average lon temperature, T1(keV)	24.1	12-0	20.0
Average electron density, u _a (10 ²⁰ /m ³)	0.81	1.50	3.39
Average feelen power, Pf(HM)	7310.	2992.	3138.
Average please power deseity, Pg/Vm	4.49	3.60	83.2
Averege mestron PV lunding, $I_{\omega}(16t/\pi^2)$	3-6	5.2	19.3
Burn-time/off-time	-	20.	

FUSION POWER	PLANT COST	COMPART BOW	(HORMA	LIZED TO THE	:)
ACCOUNT		STARI	1144	医11812.23	

	ACCOUNT	STARFIRE	医11812.23	CREPR 7
20.	Land and and land rights	0-19	_	0.38
21.	Structure and alte	20.09	12.48	19.33
22.	Booctor plant equipment (RFE)	54.00	72.04	36.04
	2 Plrot-well/blanket (PW/S)	4.77	6-66	1.71
	22.1.2 Shield (8)	10.78	4.47	-
	22-1-3 Cell (C)	9.90	12.90	3.56
	FFC - FW/8 + 8 + C	23.48	23.43	3.27
73.	Turbine plani equipment	14.47	10.68	23.49
24.	Ejectric plant equipment	6.77	3.29	14-02
25.	Riscalismesus plast equipuest	2.37	1.27	4.84
26.	Special meterials	0.014	_	0.029
90.	Direct Coote (TDC)	100.	100.	100.
99.	Total esete	185.2	200.26	173+00
	Buit direct cost, UDC (\$/kWe)	1439	2536(1685](0)	863
	Cook of electricity, COE(mille/kWe) 67.0		40.7
	Mot electric power, P.(Mie)	1200	1200	1000

3.1. In-Vecuum Componente (1VC)

Table 1II gives the neutronics response of a "typical" high-heat-flux IVC (i.e., FW) to a fusion neutror FW loading, $I_{\rm W}({\rm MW/m^2})$. Since $I_{\rm W}$ typically will be 3-5 times greater for the compact reactor ($I_{\rm W}$ = 15-20 MW/a², and even higher for the Riggetron), the radiative/conductive/convective energy fluxes emanating from the ignited DT plasma, $I_{\rm QW} \leq I_{\rm W}/4$, will be correspondingly increased for similar

⁽t)Poloidel Lield or pleams edge.

TABLE III. NEUTRONIC RESULTS FROM A "CANONICAL" COMPACT REACTOR FPC WITH PW NEUTRON LOADING L. (MW/m²)

● First-well (copper/H₂0) 14.1-MeV neutron current, J (n/m2 a) = 4.43(10) 17L Neutron flux, $\phi_{\omega}(n/m^2 e) = 4.43(10)^{18}L_{\omega}$ Total full power year fluence, $\phi_n \tau (n/m^2) =$ $1.40(10)^{26}I_{1}$ Radiation dose rate, R(rad/s) Neutrons, $R_n(rads/s) = 8.2(10)^4 L_u$ Gamma rays, $R_{v}(reds/s) = 1.3(10)^{5}I_{w}$ dpa/yr = 11L Helium appm/yr = 311. Hydrogen appm/yr = 831 Average transmutation rates Nickel $(X/yr) = 0.13I_w$ Zinc $(X/yr) = 0.11I_{\omega}$ Heat flux, I_{OW} (MW/m²) $\leq I_{w}/4$ Average power density, Q (MW/m3) = 10L a Blanket ($\Delta b = 0.6 \text{ m}$, L1-Pb/B_{Δ}C/W) Peak power density, $Q_B(MW/m^3) = 13I_w$ Average power density, $\langle Q_R \rangle$ (MW/m³) = 1.4L Average dpa/yr = 2.31, Average helium appm/yr = 26.71 Average hydrogen appm/yr = 7.71, ■ Exo-blanket coil (copper/H₂0) Peak neutron flux, $\phi_{c}(n/m^{2} s) = 3.4(10)^{16} L_{c}$ Radiation dose rate, R(red/s) Neutrons, $R_n(rads/s) = 1.2(10)^2 I_n$ Gemma rays, $R_{v}(rads/s) = 1.10(10)^{3}L_{v}$ Peak dpa/yr = 0.063L Peak helium appm/yr = 0.0271 Pack hydrogen appm/yr = 0.131 Average transmutation rates Nickel $(2/yr) = 1.1(10)^{-3}L$ Zinc $(x/yr) = 0.5(10)^{-3}L$ Peak power density, $Q_c(MW/m^3) = 0.11$

plasma conditions (i.e., profiles, edge-plasma paremeters, atc.). The power part ion between particles versus photons, as well as

the aplit of each between FW, limiter, and/or divertor, represents a crucial uncertainty for all fusion devices. The major materials questions for the IVCs are:

- Removal of both aurface (≤ L_g/4 MW/m²) and volumetric (~10L_W MW/m³) heat loads within acceptable temperature, atress, and critical-heat-flux limits, (i.e., need for materials with high thermal conductivity and high thermal atress parameter, M).
- Sputter erosion and redeposition rates for FW and limiter surfaces.
- a Long-term (swelling, creep, embrittlement, alloy charges, etc.) and short-term (thermal conductivity changes, hydrogen permeation and recycle, etc.) radiation effects.

Two limiting cases of uniform heat deposition onto IVCs can be envisaged: a) all energy is incident as radiation from a high-Zeff plasma edge or, b) all energy is convected to IVC surfaces by charge-exchange neutrals and edgeplasma particles. If all energy shed by the pleams appears as a uniform heat load, then IVC structural alloys with thermal transport properties that are better than stainless eteel will be required for the compact reactor options. Figure 3 gives the thermal stress parameter $M = 2\sigma_Y(1-\nu)/\sigma E = I_{OW}^{-\frac{1}{N}}\delta$ as a function of FW temperature; M measures the heat flux, $I_{Ou}^{-\frac{1}{1}}$, through a material of thickness δ that will cause yielding by the resulting thermal atress. For the copper-alloy and atainlesssteel materials "extrema", Fig. 4 gives the dependence of IOw allowed for a pressurizedwater-cooled tube of thickness 6 if the aum of the primary (pressure) and secondary (thermal) stress is maintained at the indicated fraction, σ/σ_V , of the yield atress; constraints relevant to electic-plastic

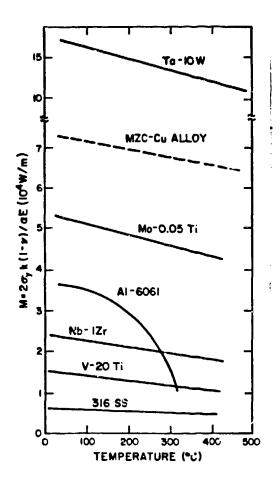


FIGURE 3

Thermal stress parameter as a function of temperature for a range of potential IVC metals.

limits, thermal ratchetting, and fatigue-creep limits, can eimilarly be applied to Fig. 4. alloy schieves performance at a lower operating temperature, which will degrade somewhat the therma' performance to an extent determined by the fraction of the fusion energy appearing in IVC coolent circuit. This important tradeoff between high-heat-flux operation, decressed FPC cost, and derated system performance remains to be comprehensively essessed in terms of a COE figure-of-merit. Indications are, however, that the significant

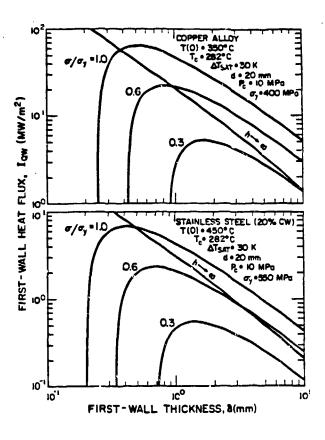


FIGURE 4

Dependence of maximum heat flux, $I_{\mathrm{OW}} \simeq I_{\mathrm{W}}/4$, allowed onto a FW coolant tube of whickness δ and cooled with pressurized water for a given primary plus accondary stress level, σ , for both atainless steel and copper alloy under the conditions indicated.

reduction in UDC accompanying the compact option reduces the COE to an extent that exceeds the increase associated with a potentially lower system performance (i.e., reduced plant efficiency, increased recirculating power fraction, and decreased plant factor).

If all the energy rejected by the plasma, on the other hand, is deposited uniformly as anergetic particles with an energy, T_E , characteristic of the plasma edge, a particle flux of $4.2(10)^{21}/T_EI_{QW}$ [~1.4(10)²³ particles/m² a for $\tau_{QW} \approx 5$ MW/m² and

Tg = 150 eV] would result. For a DT sputtering yield of ~0.02 and a FW atomic density of $\sim 8(10)^{28}$ atoms/m³ (stainless steel), gross erosion rates of > 1 m/yr would result, even if self-sputtering and ion acceleration through electrostatic sheaths were neglected. This problem is worsened if particle and energy fluxes are concentrated onto the IVC surfaces by limiter and/or divertor action. The degree to which this problem will hinder the development of fusion depends on poorly understood edge-plasma processes that are generic to magnetic fusion and not uniquely a compact reactor issue. Potential solutions to this problem are:

- Operate with edge-plasma temperatures below the sputtering threshold (< 50 eV).
- Operate with edge-plasma temperature that are well above the aputtering-yield max!mum (> 1000 eV).
- Establish a high-Z radiating plasma mantle without having the FW aupply the high-Z material through large sputtering rates.
- Design for large gross sputtering rates, but assure a nil net erosion rate through careful control of redeposition distribution.

From the viewpoint of FW survivability, these problems are not unique to or more severe for the compact reactors. Aside from differences in basic plasma processes that may result when differences of ~3-4 in average plasma density (Table II) are taken into account, the ratio of perticle flux to neutron current incident onto a FW from an ignited DT plasma should be similar for both systems, thereby decoupling somewhat the FW erosion problem from the issue of device compactness; the compact, FPC simply achieves both its neutron (dps) and erosion (mm) lifetime

"fluence" in an expected aborter chronological lifetime, but only after generating a similar total quantity of fusion energy for nominally a similar expenditure of FW/B mass. Iasues that relate specifically to device compactness and the expected higher erosion rates, however, are:

- can the compact reactor plasma survive a potentially higher recycle rate and schieve and/or remain ignited?
- depending on the heat load under which any IVC surface must function, the use of thick-walled 'ubes with an erosion margin designed to extend the sputtering life is generally less attractive for the compact systems because of the higher heat fluxes (Fig. 4).

An estimate of the effects of neutron irradiation on a copper-clloy FW, and possibly on inorganic electrical insulation if FW coils or electrical breaks are required, has been summarized in Ref. 8 and more recently for the FW copper-coil insert proposed for MARS.26 Transmutstion-induced resistivity increases in the FW copper conductor (Table III) and the dimensional stability of both the copper alloy and the proposed MgO or MgAl₂O₄ insulation²⁷ are key concerns for a FW "coil", whether actively driven (i.e., TF coil in Riggatron, H-coil in OHTE) or a passive conducting shell needed to stabilized short-wave length plasma MHD modes. Perkins²⁸ also points out that for sufficiently high voltages (> 700 V) and instantaneous radiation dose rates (> 104 Gy/s = 106 rad/e), thermal runaway through Joule heating can be potentially destructive to electrical insulators; these conditions generally apply near the FW and for relatively high-field, actively driven coils.

A increase of the electrical resistivity by radiation and transmutation effects is also accompanied by a decrease in the thermal conductivity in metals, since both current and heat are carried by electrons. A high-heatflux FW, therefore, must be designed to operate with increased thermal atress towards the end of life, although thinning of the FW by aputter erosion, if allowed, will tend to counteract the effects of decreased thermal conductivity on the FW stress. If the initially unirradiated material is a solution strengthened copper allcy, however, the decreased electrical and thermal conductivities caused by alloy additions can mask the effects of transmutation product (Ni. Zn) buildup. Although some information on radiation-induced swelling candidate inorganic insulators, similar data for copper alloy are not available at present; fission reactor irradiations of relevant alloys, however, are in progress.29 Agehardened copper alloys, such as MZC may overage or the alloying element may dissolve under irradiation; generally, 29 dispersion hardened alloys may exhibit greater radiation stability in this respect. It is noted that procedures for radiation hardening against high-energy neutrons of steering magnets for the LAMPF30 and the quadrupole beam transport magnets for FMIT³¹ have developed fabrication methods that are directly applicable to the compact fusion reactors (co-extruded Cu/Mg0 co-axial conductors with internal water cooling); the radiation fields and lifetime fluences for these accelerator applications fall short of fusion FW conditions, however. Lastly, the requirements of the FW coil proposed for the MARS design³² will satisfy the needs for most compact fusion systems. Generally, the need and potentially high payoff for high-heat-flux alloys in most IVC applications and the role

that outh alloys may play in shaping the fusion end product has only recently been recognized. 33,34

3.2. Blanket/Shield (B/S)

The B/S thickness for the compact reactor approaches is reduced to the minimum required for adequate tritium breeding and thermal energy recovery. The winimum-thickness (optimized) B/S. when coupled with the increased FW loading, schieves at least an order of magnitude increase in FPC power density, and a considerable reduction in total cost, as well as providing options for appreciably different installation maintenance achemes because of reduced FPC mass (Table II). Magnet shielding in the usual sense is not envisaged; instead a thin (0.05-0.10 m) outer region of the 0.5-0.6-mthick blanket may contain a mixture of BAC end a dense, high-Z meterial operated at the blanket temperature and cooled by the primary blanket coolent.

For FW neutron loadings in the 15-20 MW/m2 range, the local blanket rower density becomes comparable with that in the core of an LWR (> 200 MWt/m3), with the average blanket power density being in the range 30-50 MWt/u3. At the peak and average power densities envisaged for the compact reactors, ceramic breeders cooled by pressurized helium gas or water become less attractive. Because of the low lithium inventory, reduced fire hazard, and unique combination of breeder/coolant/ multiplier functions, the low-melting (235°C) lead-lithium eutectic, PbR3Li17 (referred to hereinafter as PbLi), has become a popular choice for high-power-density blankets.7.35-37

Confinement syrtems with magnetic topologies that require liquid-metal coolant to flow across magnetic fields 37,38 may be forced either to cost coolant ducts with electrical insulators 35 or to reduce the MHD

pressure drop simply by limiting the coolant flow velocity and thereby limit the FW neutron loading. 36 The high power density for the PbLi-cooled CRFPR blanket, 7,37 however, can be achieved with minimal pumping power without recourse to the use of electrically insulated coolant ducts because of the unique, low-field poloidal magnetic topology that characterizes that eyetem. The materials problems related to corrosion (particularly for ceremic coatings), tritium recovery, end tritium barriers for the compact reactors remain similar to those for other systems using similar blankets. The acceleration of stress corrosion cracking by the addition of small amounts of water to these liquid-metal systems remains as a perticularly critical concern.

Although rf and neutral-beam ducts are not envisaged for the compact systems so far considered, the task of manifolding and (vacuum) ducting appears to be more exacting. Since the gaseous (DT, He impurities) and coolent throughputs will in regnitude remain unchanged for any fusion power plant of similar power rating, the reduction of the FPC volume by at least an order of magnitude results in ducting and manifolding to regions outside the FPC becoming a more dominant part of the FPC "real estate"; FPC design integration for the compact systems becomes a more challenging exercise. 37

Lastly, even for the topologically favorable RFP, the MHD pressure drop needed to provide adequate cooling by a liquid metal to the high-heat-flux, high-power-density FW region can easily require excessive MHD pumping power. Either a ceremic coating of the FW coolent channels or a esperate pressurized-water coolent circuit will be required. The problems that attend the use of pressurized-water cooling, even in conjunction with the chemically less reactive PbLi,

presents some concern. The need to incolate/insulate thermally the lower-temperature FW coolant circuit from the higher-temperature blanket coolant circuit in order to minimize the backflow of high-quality blanket heat into the lower-quality FW heat, however, naturally results in a double, if not triple, containment of the pressurized-water coolant circuit from the liquid-metal circuit.

3.3. Magnet Coils

Most compact reactor embodiments considered to date specify water-cooled copper coils located either at or near the FW, outside the thin (0.5-0.6 m) high-power-density blanket, or both (e.g., main coils outside the blanket, feedback or current-drive coils within the blanket or at the FW). In either case, radiation-resistant in ganic electrical insulation will be required. Either insulator would be pleams-sprayed onto preformed copper conductors, or a powdered insulation (i.e., MgO or MgAl₂O₄) would be coextruded with conductor and coolant tube, the letter method being used in the fabrication of radiation-hardened coils for use in highanergy particle accelerators.30,31 Under more severe conditions, the FW coil requirement should be similar to the requirements enviseged for the MARS hybrid megnet insert, 26,32 or for the lass Lavers tokemak conditions enticipated at the in-blanket equilibrium-field coils.

The issue of coil radiation life is poorly resolved by the existing data base, but under the conditions listed on Table III, a coil at the FW location exposed to a neutron loading of $I_W = 20~{\rm MW/m^2}$ would sustain an MgAl $_2$ 0 $_4$ swalling rate of 11 volume parcent par year and a (pask) copper conductor resistivity increase of 100-200% par year. It is noted that the swelling and machanical degradation in cubic ceremics like MgO or MgAl $_2$ 0 $_4$

considerably less than axiseymetric ceramics (i.e., hexagonal Al₂0₃),²⁷ and that the increased resistivity in 300-400 K copper is related to the transmuted alloy additions rather than intrinsic point-defects. Even under fresh etertup conditions, a FW coil can significantly reduce the ownrell plant efficiency for both the OHTE19 and the Riggetron 20 reactors; operational lifetimes of only a few months are predicted for L = 20 MW/m². A strong incentive exists, therefore, to locate these coils outside the FW some and behind at least > 0.1-m of blanket. As shown in Table III, interposition of a 0.6-m-thick PbLi blankst reduces the rate of insulator awelling and conductor resistivity increase by over two orders of magnitude. Such a coil could possibly outlive the FW/B and could be recycled. Generally, however, the inceptive to move the coil outside the blankst is not driven by considerations of lifetime and the desire to reduce mean usage (1.s., operating cost), but instead by the need to: a) improve the overall plant thermal efficiency, since the FW coil would operate at a theirodynamically uninteresting temperature, b) to sese the breeding of tritium, although a few 10s of millimeters of copper has a net benefit on tritium breeding because of neutron multiplication, and c) to relieve the overall FPC congestion related to electrical/hydraulic/ thermodynamic/tritium-recovery functions. Generally, the engineering development needs from both a systems and a materials viewpoint, even for the high-field FW magnets, 19,20 should be sesier and less costly than for the large superconducting magnet designs. Lastly, a potentially significant advantage of compact systems is the facilitated use of afficient (1.e., reduced stored energy, currents, and forces) magnetic divertors because of the close proximity of magnet coils to the planma, an option available only when thin-blankated, copper-coiled compact systems are considered.

4. SUMMAKY AND CONCLUSIONS

Significant improvements in both the operational and economic prospects for fusion power are promised for systems with power densities an order of magnitude above present projections. These compact reactors will require meterials that in some areas differ from the mainline approaches.

The greatest need for materials development rests with the high-heat-flux IVCs (FW, limiters, divertors). Given that IVCs can be designed and operated with 4-5 MW/m2 heat fluxes, the critical areas reduce to the partition of radiation vareus particle flux incident upon IVC surfaces, the associated eputter erosion rate, the reposition processes (location and integrity), and the impact on the overell pleams performance of potentially large transfers of impurities around the The problems related to sputter system. erosion, however, in magnitude and kind, are not unique to compact reactors. Although sputtering rates are expected to be increased for the compect systems, given similar plasma and edge-pleame physics, the amount of FW sputtered per neutron fluence [mm/(MW yr/m2)] should be independent of the concept and simply becomes a matter of "fluence".

Hence, the potentially unique materials problems for compact systems are related to the need to understand and control the bulk mechanical radiation demage properties of the new FW materials (copper, vanadium, molybdenum slloys) required to deal with the increased heat fluxes. Even then, such materials may be used in pumped limiters and/or divertor plates for the larger superconducting fusion systems.

The compact reactor option narrows the many B/S choices listed in Ref. 2 to a few concepts that can operate at local and average power densities considered aconomically necessary for other nuclear power systems (Fig. 1). The magnet development required to produce relatively small, radiation—hardened resistive coils appears to be well advanced, 30,31 albeit on a reduced scale. Hence, for both B/S and magnet areas, the materials requirements for the compact options appear no more difficult, and in many respects easier, then the mainline program needs.

In aummery, all materials issues for compact reactors are being or can be addressed within the mainline program. A new emphasis, however, must be placed on understanding, creep. fatigue, fatigue-creep interaction, alloy stability, coolent-alloy interaction, atc. for these new high-heat-flux systems. It is this classical area of materials and systems engineering, as applied to IVC surfaces, that major strides can be made in adv: ing fusion as a truly compatitive energy source.

ACRONYMS

- B/S Blenket and Shield
- COE Cost of Electricity (mills/kWsh)
- FPC Fusion Power Core (FW. B/S. and coils)
- FW First Well
- ADC Total Direct Cost
- RPE Reactor Plant Equipment (Account 22) cost
- UDC Unit Direct Cost (\$/kWs)
- TFC Toroidel-Field Coil
- PFC Foloidel-Field Coil
- OHC Ohmic-Hesting Coil
- EFC Equilibrium-Field Coil

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